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| Journal Name | Research on Chemical Intermediates | |
|--------------|------------------------------------|--|

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| Schedule | Received | 19 October 2020 |
| | Revised | |
| | Accepted | 30 October 2020 |

Abstract Porous carbons are materials of wide application and their request is more and more increasing in the last years: Properly designed synthesis is presently available for the preparation of materials to be used in the several fields (e.g.: adsorption, molecular separation, and catalysis). The characterization of the porous carbons is usually carried out using different techniques such as thermogravimetric analyses, Raman spectroscopy, Scanning electron microscopy, etc. In this work, the micro-Raman technique is adopted in

combination with N₂ physisorption at 77 K to monitor how the synthetic approach influences the presence of either amorphous or ordered regions in porous carbons. The typical D and G Raman bands of activated carbons have been carefully deconvoluted in six different components by a fitting procedure, and the determined $R_1 = I_{D1}/I_G$ ratio correlated to their specific surface area.

Keywords (separated by '-') Hyper-crosslinked polymers - Micro-raman spectroscopy - N₂ physisorption - SSA-BET

Footnote Information **Electronic supplementary material** The online version of this article (<https://doi.org/10.1007/s11164-020-04338-x>) contains supplementary material, which is available to authorized users.



1 On the correlation between Raman spectra 2 and structural properties of activated carbons derived 3 by hyper-crosslinked polymers

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5 Received: 19 October 2020 / Accepted: 30 October 2020
6 © Springer Nature B.V. 2020

7 Abstract

8 Porous carbons are materials of wide application and their request is more and more
9 increasing in the last years: Properly designed synthesis is presently available for the
10 preparation of materials to be used in the several fields (e.g.: adsorption, molecular
11 separation, and catalysis). The characterization of the porous carbons is usually car-**AQ1**
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14 que is adopted in combination with N₂ physisorption at 77 K to monitor how the
15 synthetic approach influences the presence of either amorphous or ordered regions
16 in porous carbons. The typical D and G Raman bands of activated carbons have
17 been carefully deconvoluted in six different components by a fitting procedure, and
18 the determined R₁ = I_{D1}/I_G ratio correlated to their specific surface area.

19 **Keywords** Hyper-crosslinked polymers · Micro-raman spectroscopy · N₂
20 physisorption · SSA-BET

21 Introduction

22 In the last years, the demand of new porous materials has rapidly increased in
23 relation to the application in different fields, such as air and water decontamina-
24 tion, gas and fuel storages, molecular separation, catalysis, etc. [1–4]. Syntheses

A1 **Electronic supplementary material** The online version of this article (<https://doi.org/10.1007/s1116>
A2 [4-020-04338-x](https://doi.org/10.1007/s11164-020-04338-x)) contains supplementary material, which is available to authorized users.

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25 of porous materials are designed in order to obtain materials that satisfy specific
26 requirements for their final application. In particular, desired physical character-
27 istics as high specific surface area, pore dimensions and volume are essential for
28 adsorption [5–8], molecular separation [9–11] and catalysis [12–14]. Nowadays,
29 various classes of porous materials with different chemical compositions have
30 been developed, among which inorganic zeolites and synthetic clays (bentonite,
31 montmorillonite), organic–inorganic hybrid frameworks (e.g., metal–organic
32 frameworks MOFs) [15–17], covalent-organic frameworks (COFs) possibly with
33 heteroatoms [18], porous organic polymer (POP) [19, 20] and activated carbons
34 (ACs) [21–25].

35 The POP includes different types of material, and among these the porous aromatic
36 frameworks (PAFs) and hyper-crosslinked polymers (HCPs) families have
37 received great attention in relation to their peculiar properties such as high pore
38 volume (tunable from micro- to meso-porosity) and physico-chemical stability
39 (especially in aqueous environment) and shows the possibility to perform com-
40 positional control during their synthesis: using different linkers it is possible to
41 obtain a large number of different porous structures and modulate the hierarchical
42 porosity [26–36].

43 A wide family of porous materials that has been investigated for decades is
44 that of carbonaceous solids, obtained by carbonization of biomass [37–39],
45 though the use of raw materials with not constant chemical composition leads
46 to the preparation of activated carbon whose physico-chemical properties are not
47 completely controllable.

48 One strategy to overcome this problem is the use of polymers as raw materials
49 [40], and adopting selected physical or chemical activations [41, 42]. In addi-
50 tion, these materials are very simple to synthesize, and the costs of production are
51 lower than other materials [43].

52 Many examples of this strategy, such as the use of HCPs for the preparation of
53 microporous carbonaceous materials, can be found in the recent literature [44].

54 A key point is the subsequent phase of characterization to have a clear under-
55 standing of the properties of the materials, and it is not straightforward to relate
56 the information obtained from different techniques. A correlation between
57 Raman analyses and the porosimetric properties of carbons could be a valid and
58 rapid alternative to the use of other techniques and be used as a methodological
59 approach in the selection of the most promising materials. Moreover, the Raman
60 technique does not need any particular manipulation of the samples, avoiding
61 material modifications consequent to the preparation steps prior the analyses.

62 Using this approach, we have performed micro-Raman spectroscopy analyses
63 for the characterization of porous carbons obtained by HCPs prepared with dif-
64 ferent ratios of precursors (tetraphenylmethane (TPM) and formaldehyde dime-
65 thyl acetal (FDA)), and subsequently carbonized in three different ways [44]. The
66 Raman spectra are elaborated with a specific fitting deconvolution process, aim-
67 ing at acquiring information on the material structures, in order to assess the level
68 of amorphous vs crystalline components. These data are subsequently paired with
69 textural properties (e.g., specific surface area) of studied materials.

70 Materials and methods

71 Synthesis of HCP materials (UPO) and post-synthesis treatments (KUPO)

72 The HCP materials have been synthesized by using procedure already described
 73 elsewhere [44]. The hyper-crosslinked polymers (UPO) were prepared using fer-
 74 ric chloride (Sigma Aldrich, quality level MQ 200) and tetraphenylmethane (TPM
 75 97%; Sigma-Aldrich) suspended in dichloroethane (DCE 99.8%; Sigma-Aldrich),
 76 stirred at room temperature until a homogeneous thick gel was obtained.

77 Formaldehydedyethylacetal (FDA, Sigma-Aldrich, quality level MQ 200)
 78 was added drop-wise, to the thick gel under stirring, then heated under reflux,
 79 diluted in ethanol after cooling to room temperature, washed until neutral pH, and
 80 finally dried in oven at 100 °C. The reaction was carried out using two different
 81 TPM:FDA molar ratios of 1:8 and 1:16, respectively.

82 Different procedures of chemical activation based on thermal and chemical
 83 treatments have been subsequently used to modify the contact between the acti-
 84 vating agent (KOH) and the polymers matrix. In particular, three different activa-
 85 tion procedures were followed: (1) in the pre-activation method 1 a solid solution
 86 of UPO material and KOH (ratio 1:3) is prepared under inert conditions; (2) in
 87 the pre-activation method 2 a solid solution of UPO and KOH (ratio 1:3) is pre-
 88 pared in air for 5 h in order to allow ambient water hydrating KOH thus facilitat-
 89 ing dissolution into the UPO material and (3) in the pre-activation method 3 UPO
 90 material is stirred in a solution of KOH (1 M) in ethanol for 18 h, and subse-
 91 quently dried under vacuum at 50 °C.

92 After the preparation, these mixtures were heated to 750–800 °C, and subse-
 93 quently hold under isothermal conditions for 1 h. The carbonaceous materials
 94 obtained by these procedures were labeled as AC x - y - z , where x refers to the pre-
 95 activation method, y indicates the TPM:FDA ratio of the parent polymer and z
 96 the activation temperature. The obtained samples are listed in Table 1.

Table 1 Synthesis procedure of the samples analyzed in this work

| Sample name | Pre-activation method | TPM:FDA molar ratio | Heating (°C) |
|-------------|--------------------------|---------------------|--------------|
| AC1-8-750 | KOH (1:3) inert | 1:8 | 750 |
| AC1-8-800 | KOH (1:3) inert | 1:8 | 800 |
| AC1-16-800 | KOH (1:3) inert | 1:16 | 800 |
| AC2-8-750 | KOH (1:3) air | 1:8 | 750 |
| AC2-8-800 | KOH (1:3) air | 1:8 | 800 |
| AC2-16-800 | KOH (1:3) air | 1:16 | 800 |
| AC3-8-800 | KOH (1 M) ethanol (18 h) | 1:8 | 800 |
| AC3-16-750 | KOH (1 M) ethanol (18 h) | 1:16 | 750 |

97 Characterization techniques

98 The synthesized powders were analyzed by micro-Raman spectroscopy, using a
99 JobinYvon HR800 LabRam μ -spectrometer (HORIBA JobinYvon, Paris, France).
100 The spectra were collected at room temperature, addressing the laser beam onto
101 the samples thanks to the annexed Optical Microscope (OM; Olympus BX41;
102 Olympus, Tokyo, Japan). The instrument was equipped with a HeNe 20 mW laser
103 (wavelength 632.8 nm) and a charge-coupled air-cooled device detector. An instru-
104 ment calibration was performed checking the position and intensity of the Si band
105 at $520.65 \pm 0.05 \text{ cm}^{-1}$ before running each experimental acquisition. In order to
106 obtain the best signal/noise ratio, every spectrum was obtained carrying out 30
107 exposures of 120 s for the 16 spectral ranges that cover the $3500\text{--}750 \text{ cm}^{-1}$ (about
108 16 h) and where characteristic first- and second-order bands of the carbonaceous
109 materials (CMs) lie [45–49]. No filter was applied to the signal for spectra acquisi-
110 tions, after the checking that no sample modification was observed addressing a full
111 power beam. Spectra were also acquired by a JobinYvon Evolution micro-Raman
112 spectrometer (HORIBA JobinYvon, Paris, France), equipped with a Nd:YAG laser
113 source (wavelength 532 nm). Autocalibration of the instrument was carried out
114 before every experimental session, using the $\sim 520.6 \text{ cm}^{-1}$ band of Si. The spectra
115 were acquired addressing 3.2% of the laser power on the materials, with 2 exposures
116 of 30 s: this experimental setting allowed obtaining good spectra and no sample
117 modifications under laser energy.

118 In order to exclude sample modification under green laser, different times and
119 excitation source powers have been applied to carry out sample characterization.
120 Applying a longer time or a higher power, it was evident sample modification during
121 analyses, observing both Raman patterns and optical modifications of the samples
122 after spectral acquisitions.

123 A fitting procedure was performed using the package software of the OPUS V.
124 5.5 on the first-order CM bands, with different runs in order to obtain the minimal
125 residual error, considering six different components (five Lorentzian bands, G, D1,
126 D2, D4, and D5 one Gaussian band, D3) after baseline subtraction of the fluores-
127 cence effect. The obtained results were utilized to calibrate an automatic 6-way fit-
128 ting process using Fityk V. 0.9.8 software in order to maintain the same operational
129 procedure in spectrum elaborations and minimize artificial variations. This process
130 was set using a compiled script aimed at the automatic recognition of five Lorentz-
131 zian and one Gaussian (D3) bands. The script firstly fits D and G bands, then searches
132 at the same time D2 and D3 bands and finally recognizes the last two bands centered
133 at about 1270 and 1150 cm^{-1} (D4 and D5 bands). All the obtained results were com-
134 pared with the manual fitting process performed using OPUS software.

135 The N_2 physisorption measurements were carried out at 77 K in the relative pres-
136 sure (P/P0) range 1×10^{-7} –1 by using an Autosorb iQ/ASiQwin instrument (Anton
137 Paar QuantaTec Inc., Boynton Beach, FL, USA). Prior to the analysis, the samples
138 were outgassed at $150 \text{ }^\circ\text{C}$ for 16 h (residual pressure lower than 10^{-6} Torr). The
139 apparent BET surface areas were calculated over a relative pressure range rec-
140 ommended by the “Micropore BET Assistant”, a program that is implemented in
141 ASiQWin Quantachrome software to facilitate the selection points within the linear

142 range of the BET plot for microporous materials. The pore size distribution for all
 143 samples is calculated using quenched solid density functional theory (QSDFT) on
 144 a carbon surface with slit/cylindrical geometry applied to the adsorption branch
 145 (method with the smallest fitting error).

146 Results

147 Micro-Raman spectroscopy was applied to the activated carbons (ACs) synthesized
 148 from hyper-crosslinked polymers (HCPs) reported in Table 1. The characteristics D
 149 and G bands, each of them being composite of several components, were always
 150 observed (Fig. 1).

151 Deconvolution process was performed to extract the single components, which
 152 were tentatively correlated to the specific surface area of the synthesized samples.
 153 An automatic six-way fitting applied by means of the Fityk software was optimized
 154 in order to determine the ratio between the intensity of D1 (disorder band) and G
 155 (graphitic band) (R_1), obtained from the spectra acquired by the two laser sources
 156 as proposed by Beyssac et al. in 2002. [46]. A third fully automated method (IFORS
 157 software) was applied to further validate the manual and the semiautomatic fitting
 158 processes (data not shown), in particular to check the position of the baseline used
 159 for background subtraction and peak positions.

160 On the synthesized materials, by using red and green laser sources two different
 161 shapes of the first-order Raman bands were observed. Figure 1a refers to the data
 162 obtained with the laser at 632.8 nm, whereas Fig. 1b to spectra acquired with the
 163 laser at 532 nm wavelength.

164 Finally, two examples of first-order band fitting are reported in Fig. 2a, b high-
 165 lighting peak position and band labels.

166 The D1 band at about 1330 cm^{-1} is due to defective sp^3 carbon: The intensity
 167 and position of this band depend on the presence of amorphous regions in the
 168 carbon materials. Conversely, the intensity of bands at $1550\text{--}1580\text{ cm}^{-1}$ (G band)

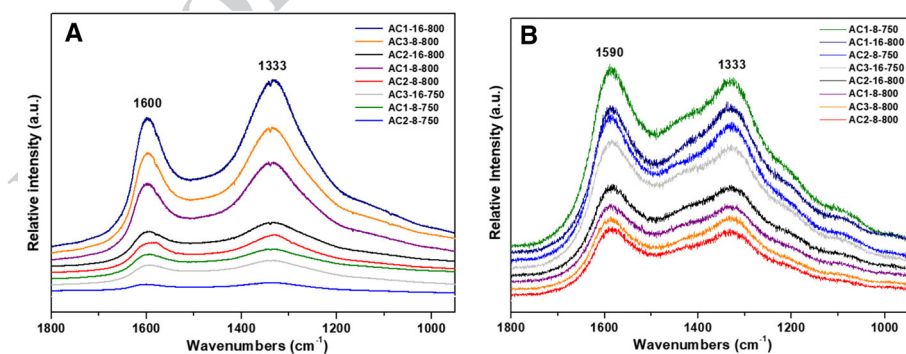


Fig. 1 Representative Raman spectra obtained on ACs samples. **a** Raman spectra obtained by 632.8 nm laser source; **b** Raman spectra obtained by 532 nm laser source

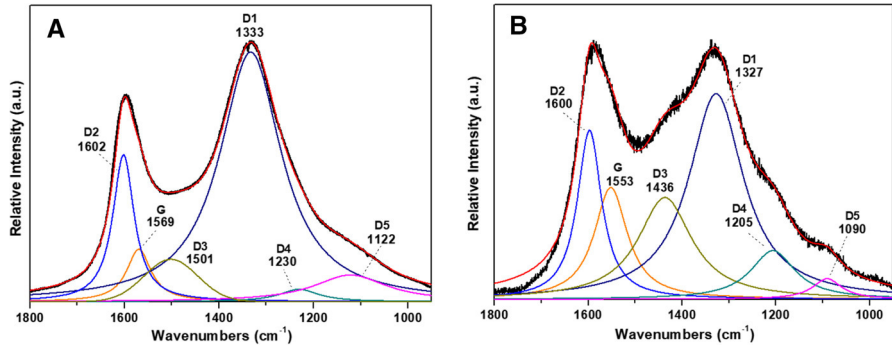


Fig. 2 Example of spectrum fitting of the sample AC1-16-800 with 632.8 nm laser source **a** and 532 nm laser source; **b**; above the band position values, the labels of the six obtained bands are reported

169 is related to the extension of ordered regions [50]. The D1 band is strictly related
 170 to that lying between 1600 and 1615 cm^{-1} , called D2 band.

171 These bands are relevant to estimate the disorder degree in carbonaceous struc-
 172 tures, and accurate best fitting evaluations are needed to obtain reliable informa-
 173 tion. In addition, other three bands lie in the 1500–1000 cm^{-1} range: the band
 174 at around 1500 cm^{-1} is reported in the literature as D3, while the band at about
 175 1230–1200 cm^{-1} is called D4, and the last at about 1100–1120 cm^{-1} is called D5.

176 Different numbers of bands in this spectral range are reported in the literature,
 177 and this confirms that the study of carbon materials using Raman features is not
 178 yet univocally made. For the sake of simplicity and clarity the bands nomencla-
 179 ture is reported onward in this work [45–61].

180 Automatic fitting method was validated comparing the obtained R_1 ratios
 181 where the R_1 is the intensity ratio between D1 and G bands:

182

$$R_1 = \frac{I_D}{I_G} \tag{1}$$

183

184 Small R_1 values indicate that the material is characterized by a greater struc-
 185 tural order: When the carbon is more crystalline, the G band shows higher inten-
 186 sity in comparison to the D1.

187 Data obtained by the fitting process and the calculated R_1 ratios are reported in
 188 Tables 2 (red laser source) and 3 (green laser source).

189 The R_1 obtained by the two laser sources were related to the values of SSA
 190 [44] obtained by physisorption analysis of nitrogen at 77 K (Fig. 3, and Tables 2
 191 and 3).

192 The highest R_1 values are obtained on samples AC1-16-800, AC3-8-800, AC1-
 193 8-800 and AC2-16-800 using red laser source. This trend is confirmed by the data
 194 obtained using green laser. The first two samples show also the highest values of
 195 V_{Total} and V_{micro} , while V_{meso} values are low ($< 0.3 \text{ cm}^3/\text{g}$). The AC-3-8-800 sample
 196 has a high V_{Total} , but the lowest SSA ($< 2400 \text{ m}^2/\text{g}$) and V_{micro} ($< 0.7 \text{ cm}^3/\text{g}$), and
 197 also the highest V_{meso} among the studied samples ($> 0.5 \text{ cm}^3/\text{g}$). The samples AC1-
 198 8-800 and AC2-16-800, as AC1-16-800, have high SSA and V_{micro} and low V_{meso} .

On the correlation between Raman spectra and structural...

Table 2 Band parameters obtained from Raman spectra acquired with 632.8 nm laser source applying the automatic fitting procedure described in the experimental section

| Sample | R_1^a | R_2^b | FWHM ^c D1 [cm ⁻¹] | FWHM ^c D2 [cm ⁻¹] | Pos. ^d D [cm ⁻¹] | Pos. ^d G [cm ⁻¹] | Diff. G-D [cm ⁻¹] | SSA [m ² g ⁻¹] |
|------------|---------|---------|---|---|---|---|-------------------------------|---------------------------------------|
| AC1-8-750 | 1.88 | 0.78 | 119.78 | 46.01 | 1340 | 1576 | 236 | 2562 |
| AC1-8-800 | 2.30 | 0.74 | 133.23 | 43.44 | 1341 | 1583 | 242 | 2666 |
| AC1-16-800 | 2.39 | 0.76 | 148.49 | 43.02 | 1337 | 1577 | 240 | 2950 |
| AC2-8-750 | 1.79 | 0.65 | 142.68 | 47.06 | 1336 | 1589 | 253 | 2500 |
| AC2-8-800 | 1.60 | 0.68 | 118.64 | 42.85 | 1333 | 1579 | 246 | 2400 |
| AC2-16-800 | 2.25 | 0.70 | 127.33 | 48.28 | 1340 | 1578 | 238 | 2700 |
| AC3-8-800 | 2.35 | 0.75 | 134.69 | 42.59 | 1340 | 1582 | 242 | 2975 |
| AC3-16-750 | 2.38 | 0.66 | 145.28 | 59.92 | 1345 | 1560 | 215 | 2318 |

^a $R_1 = I_{D1}/I_G$

^b $R_2 = \text{area [D1/(G+D1+D2)]}$

^cFWHM=Full Width at Half Maximum

^dPos. = Band position

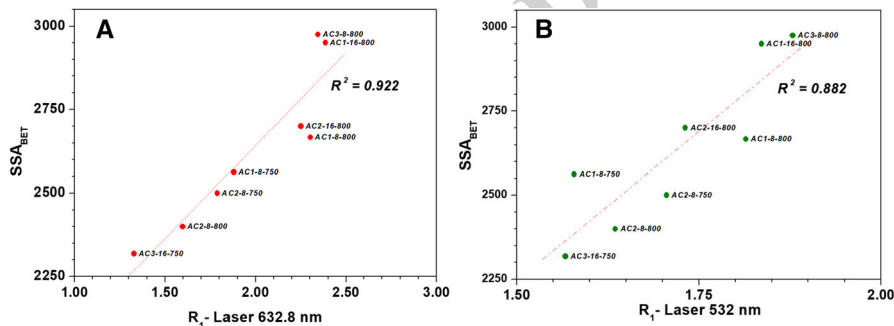


Fig. 3 a R_1 ratio obtained by red laser source versus SSA-BET data; b R_1 ratio obtained by green laser source versus SSA-BET data

199 As discussed in the following section, the obtained data show a linear correlation
200 between R_1 and SSA.

201 **Discussion**

202 During the last decades, it has been demonstrated that micro-Raman spectroscopy
203 is a powerful tool in ACs characterization, in particular for the order/disorder and
204 defects definition. Raman spectra of ACs are, at first sight, defined by a peculiar

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Table 3 Band parameters obtained from Raman spectra acquired with 532 nm laser source applying the automatic fitting procedure described in the experimental section

| Sample | R_1^a | R_2^b | FWHM ^c D1 [cm ⁻¹] | FWHM ^c D2 [cm ⁻¹] | Pos. ^d D [cm ⁻¹] | Pos. ^d G [cm ⁻¹] | Diff. G-D [cm ⁻¹] | SSA [m ² g ⁻¹] |
|----------------|---------|---------|---|--|--|--|-------------------------------------|---------------------------------------|
| AC1-8-750 | 1.58 | 0.57 | 139.22 | 62.08 | 1327 | 1553 | 226 | 2562 |
| AC1-8-800 | 1.81 | 0.58 | 140.43 | 64.66 | 1327 | 1549 | 222 | 2666 |
| AC1-16- 800 | 1.84 | 0.59 | 141.61 | 62.84 | 1327 | 1552 | 225 | 2950 |
| AC2-8-750 | 1.70 | 0.58 | 137.39 | 60.25 | 1327 | 1553 | 226 | 2500 |
| AC2-8-800 | 1.63 | 0.59 | 143.84 | 61.06 | 1328 | 1555 | 227 | 2400 |
| AC2-16- 800 | 1.73 | 0.58 | 142.24 | 63.99 | 1325 | 1550 | 225 | 2700 |
| AC3-8-800 | 1.89 | 0.59 | 131.71 | 59.22 | 1327 | 1552 | 225 | 2975 |
| AC3-16- 750 | 1.57 | 0.56 | 141.99 | 63.60 | 1325 | 1551 | 226 | 2318 |

$$^a R_1 = I_{D1}/I_G$$

$$^b R_2 = \text{area} [D1/(G + D1 + D2)]$$

^cFWHM = Full Width at Half Maximum

^dPos. = Band position

205 spectral feature, formed by first- and second-order vibrations [46–49]. The intensity
206 ratio among these bands, and in particular the one between D and G intensities (R_1)
207 is an important parameter related to the carbon organization in the samples [57].

208 The works in literature proved that 514 or 532 nm wavelengths give the best sig-
209 nal/noise ratio in ACs characterization [54]. In fact, these laser sources give Raman
210 spectra with relative low fluorescence effects in the range where the ACs bands
211 lie, allowing band fitting with minor errors in the obtained band parameters than
212 from results derived by red lasers. Nevertheless, the excitation energy of green laser
213 sources is generally more powerful, and they might modify ACs during the analyses.
214 For this reason, in this work, two different laser sources were used in AC characteri-
215 zation for the determination of spectroscopic parameters.

216 Several works studied adopted different best fitting procedures to deconvolute the
217 components of the two more intense bands constantly present in all AC spectra and
218 proposed a series of equations to calculate disorder/order ratios [46, 48, 51, 55, 56].
219 The different approaches extracted a variable number of components as a result of
220 the fitting process, thus providing a great variability in band names and mathemati-
221 cal methods. In Table 4 and SI1 the band labelling in several referenced papers are
222 listed. One of the most crucial variables is the number of components by which the
223 deconvolution process is made; a second one is the type of fitted band (e.g.: Gauss-
224 ian, Lorentzian, Voigt) [59, 62].

225 From the data reported in Tables 4 and SI1 it is clear that the number and name of
226 the fitted bands vary from one study to another. In this work the best fitting process,
227 i.e., that with the minimum residual error, adopted a six-way fitting and the decon-
228 voluted bands were named G, D1, D2, D3, D4, and D5. The R_1 values obtained from

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Author Proof

Table 4 List of Raman studies of carbonaceous materials where the bands in the 1800–900 cm⁻¹ spectral range have been deconvoluted and classified

| Laser wavelength | Name and position (cm ⁻¹) of the C bands | | | | | | Reference |
|-----------------------|--|--------------------|--------------------------|------------------|-----------------------|---------------------|-----------|
| | G | D1 | D2 | D3 | D4 | D5 | |
| 514 nm + 532 nm | G | D1 | D2 | D3 | - | - | [46] |
| 514, 633, 780 nm | G~1580 | D1 (D)~1350 | D2 (D')~1620 | D3 (D'"; A)-1500 | D4 (D)~1200 | - | [49] |
| 514 nm | G | D1 | D2 | D3 | D4 | - | [62] |
| 488 nm | G~1580 | D1~1350 | D2~1620 | D3~1500 | D4~1250 | - | [59] |
| 457, 514, 633, 752 nm | G | D | D' | D3 | D4 | - | [51] |
| 266, 532, 633, 785 nm | G~1580 | Dvariable by laser | D'variable by laser | A~1530 | TPA variable by laser | - | [55] |
| 532 nm | G~1585 | D~1350 | D'not indicated, ~1610 | D'1500–1550 | D*1150–1200 | - | [53] |
| 514, 632 nm | G | D | D' | D3~1438 | D2~1260 | D1~1185 | [60] |
| 532 nm | G~1585 | D~1350 | D'not indicated | D'1500–1550 | D*1150–1200 | - | [58] |
| 532 nm | G | D1 | D2 | D3 | D4 | - | [61] |
| 473 nm | G1610-1570 | D1350 | G _R 1500-1570 | D31450-1500 | S1100-1200 | S _L 1250 | [52] |
| - | G | D | D' | D3 | D4 | D5 | [45] |
| 532, 632.8 nm | G | D1 | D2 | D3 | D4 | D5 | This work |

The data obtained in this work are reported for comparison

* a band at 1700 is assigned to the presence of C=O groups

229 the fitted first-order bands, when compared to the SSA data, provided a first impor-
230 tant information about the disorder degree of the ACs structures. In fact, R_1 shows
231 medium–high values when the measured SSA are medium–high (Table SII): despite
232 the limited variability of these parameters, and their completely different experimen-
233 tal nature, a linear relationship between R_1 and SSA can be inferred. The reason for
234 this is that both parameters are correlated with the structural/morphological nature
235 of the carbons under study. High R_1 values are related, in fact, to a more extended
236 disorder inside the AC materials and it is highly possible that disordered carbons
237 display high specific surface area (SSA), as proposed by Roh et al. [63] on Activated
238 Carbon Fibers (ACFs).

239 Finally, as far as the polymer carbonization at high temperature is con-
240 cerned, it should be noticed that it occurs in the presence of KOH activation
241 agent as a solid–solid/solid–liquid reaction [64], according to the general Eq. 6
242 $\text{KOH} + 2\text{C} \rightarrow 2\text{K} + 3\text{H}_2 + 2\text{K}_2\text{CO}_3$. Moreover, when the activation temperature
243 increases at above 700 °C, the K_2CO_3 produced by the reaction starts to decom-
244 pose into K_2O and CO_2 . The high microporosity in the carbon matrix is formed due
245 to the triple effects of chemical activation, physical activation, and the expansion
246 of the carbon lattices by metallic potassium intercalation: All these effects lead to
247 highly porous carbons with very high SSA.

248 Conclusions

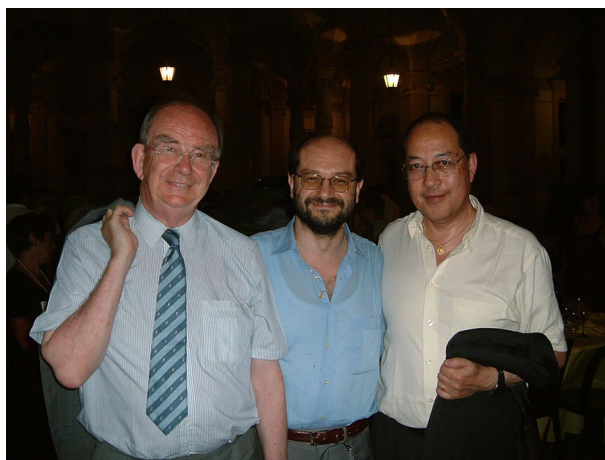
249 From the obtained results, it is found that micro-Raman data, considering the whole
250 series of parameters obtained by fitting process of the first-order bands, are related to
251 specific surface area of AC materials. In particular, among the considered calculated
252 Raman parameters, a linear relationship between R_1 (I_{D1}/I_G) and SSA-BET values
253 can be proposed. The correlation has been proved using as excitation sources for AQ2
254 Raman spectra both 632.8 and 532 nm laser wavelengths, and this suggests that not
255 significant structural variations occur during the prolonged exposure (16 h) under
256 the red laser source of samples here investigated. This result on porous carbons con- AQ3
257 firms the trend observed in the work of Roh et al., 2008, on activated carbon fibers
258 [63]. Further analyses should be carried out on a larger statistical significant number
259 of samples to understand if this correlation may be predictive of physical charac-
260 teristics (especially SSA) of synthesized porous carbons, trying to calibrate Raman
261 spectroscopy as a method for this kind of studies.

262 A memory of Prof. Michel Che

263 I met Prof. Michel Che for the first time in Paris on 16 November 1988, when I was
264 a Ph.D. student under the supervision of Prof. Salvatore (Totò) Coluccia, one of his
265 best friends. I do remember clearly that moment because it was the first birthday of AQ4
266 my daughter Isabella, and I went to Paris with Totò and Prof. Adriano Zecchina for
267 the first meeting of a European project coordinated by Prof. Che. The memory of
268 that occasion is vivid, I collected the last IR spectra of CO adsorbed on Pt dispersed

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269 on TiO₂, just one hour before catching the TGV for Paris, and during the travel
270 I completed the last slides. The excitement was very high because it was my first
271 experience abroad as PhD and I had to speak in front of famous persons, just known
272 reading the literature, Michel Che and Geoffrey Bond. Since that meeting I appreci-
273 ated the human qualities of Michel, as I used to name him more familiarly, his love
274 for science and his care for young scientists. During the last 30 years I met him sev-
275 eral times, and I have been always fascinated by his capacity of dialogue on life and
276 science, great remembrances are his conversations with Totò and sir John Meurig
277 Thomas, always ironic and full of jokes, beside relevant scientific issues. The legacy
278 of Michel on surface chemistry is invaluable, his open mind and humanity will live
279 forever in all people who had the pleasure to know him.



280
281 Photo: with Prof. Michel Che (right end), me (Prof. Leonardo Marchese) (center),
282 and Prof. Sir John M. Thomas (left end) in Torino in 2012.

283 **Acknowledgements** The financial support of the Project SATURNO "Piedmont regional operational pro-
284 gram (FESR 2014/2020) technological platform " (Bioeconomy I.1b.2.2_14/20_Bioeconomics—Project
285 code 333-19) is acknowledged.

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