

The $P2_1/m$ - $C2/m$ phase transition in synthetic amphiboles in the system $\text{Li}_2\text{O-Na}_2\text{O-MgO-SiO}_2\text{-H}_2\text{O}$: an high- T FTIR study

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Abstract: The $P2_1/m \leftrightarrow C2/m$ phase-transition has been studied by high- T FTIR analysis on a series of synthetic amphiboles in the $\text{Li}_2\text{O-Na}_2\text{O-MgO-SiO}_2\text{-H}_2\text{O}$ (LNMSH) system. Spectra were collected in the T range 25–450 °C on KBr disks. All examined amphiboles have $P2_1/m$ symmetry at room T . Their OH-stretching FTIR spectrum consists of two main bands at ~ 3740 and 3715 cm^{-1} . At the transition temperature (T_c), these bands merge into one single absorption centred at $\sim 3720 \text{ cm}^{-1}$, and no further change is observed beyond this T . Significant modifications consisting in peak shifting and band broadening are also observed in the MIR (medium infrared) $1300\text{--}640 \text{ cm}^{-1}$ region. T_c values for the different compositions were estimated based on various methods; the most reliable procedure is considered to be the fit of Landau 2-4-6 potentials using band shifts observed in the MIR region. The T_c values obtained for all samples are consistent with previous results obtained on two members of the series examined here by single-crystal or synchrotron powder HT-XRD (high- T X-ray diffraction). They correlate linearly with the aggregate cation radius at $M(6)$ [T_c (°C) = $803 - 533 \langle r_{M(6)} \rangle$; $R^2 = 0.97$]. This work thus provide a measure of the role played by the size of the $M(6)$ polyhedron in determining the T_c in simple chemical systems where the B-site occupancy (and geometry) is the only variable. The slope of the equation is far less steep in the LNMSH system than in cummingtonite; crystal-chemical reasons for this behaviour are discussed, and the local order between A and monovalent B cations is suggested to be the major constraint. In more complex systems, inspection of the available data shows that other factors such as the aggregate size of the strip of octahedra must be taken into account.

Key-words: synthetic amphiboles, LNMSH system, HT-FTIR spectroscopy, phase transition.

1 Introduction

2 Monoclinic amphiboles with small B cations (Mg, Fe^{2+})
3 undergo a displacive $P2_1/m \leftrightarrow C2/m$ phase-transition as
4 a function of temperature (T) and/or pressure (P), and the
5 T_c and P_c values depend on composition and/or cation order,
6 which may be followed by the aggregate cation radius at the $M(6)$ site,
7 $\langle r_{M(6)} \rangle$ (Yang & Hirschmann, 1995; Yang & Prewitt, 2000; Boffa Ballaran
8 *et al.*, 2000, 2001, 2004). More recently, the same transition behaviour was
9 detected in synthetic amphiboles with 1:1 monovalent and small divalent B cations
10 (*e.g.*, $^{\text{B}}(\text{NaMg})$ and $^{\text{B}}(\text{LiMg})$), and evidences were found that T_c depends on the aggregate
11 cationic radius at both the $M(6)$ and $M(1, 2, 3)$ sites (Cámara *et al.*, 2003a, 2008; Iezzi
12 *et al.*, 2004, 2005; Welch *et al.*, 2007). Notwithstanding these efforts, a comprehensive
13 P - T - X model for the phase transition in amphiboles still needs to be developed.

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18 Oberti *et al.* (2000) used single-crystal X-ray refinement (SREF), Secondary Ion Mass Spectrometry (SIMS), and
19 Fourier Transform Infrared spectroscopy (FTIR) to show that some synthetic amphiboles obtained by Gibbs
20 *et al.* (1962) and Maresch & Langer (1976) in the $\text{Li}_2\text{O-Na}_2\text{O-MgO-SiO}_2\text{-H}_2\text{O}$ (LNMSH) system have a very peculiar
21 mixture of B cations ($\text{Mg}_{0.97}\text{Li}_{0.27}\text{H}_{0.12}\text{Na}_{0.64}$) and $P2_1/m$ symmetry. This finding prompted a series of experimental
22 studies aimed at characterizing the actual composition and structural features of LNMSH amphiboles under different
23 T , P and X conditions.

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29 Cámara *et al.* (2003a) and Iezzi *et al.* (2004) showed by single-crystal XRD analysis that the Li-free
30 $^{\text{A}}\text{Na}_{0.81}^{\text{B}}(\text{Na}_{0.81}\text{Mg}_{1.19})^{\text{C}}\text{Mg}_5^{\text{T}}\text{Si}_8\text{O}_{22}^{\text{W}}(\text{OH})_2$ composition (sample 334) has $P2_1/m$ symmetry at room T , and undergoes
31 a second-order $P2_1/m \leftrightarrow C2/m$ phase transition at $T_c = 257$ °C. Iezzi *et al.* (2005) found by synchrotron
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radiation XRPD (X-ray powder diffraction) that the stoichiometric ${}^A\text{Na} {}^B(\text{LiMg}) {}^C\text{Mg}_5^T\text{Si}_8\text{O}_{22}^W(\text{OH})_2$ composition has a T_c of 326 °C. Also, a treatment of the data in the frame of the Landau theory allowed these authors to show that the thermodynamic character of the phase transition is tricritical for this composition. It is thus interesting to check the transition behaviour within the ${}^B\text{Na} \rightarrow {}^B\text{Li}$ solid solution. Iezzi *et al.* (2006) synthesized a series of terms in this solid solution, and showed by TEM and FTIR that they all have $P2_1/m$ symmetry at room T . In this work, we examine by *in situ* high- T FTIR spectroscopy the complete set of samples of Iezzi *et al.* (2006) with the aim to investigate changes in the T_c of the $P2_1/m \leftrightarrow C2/m$ phase transition. Noteworthy, the LMNSH system does not allow order-disorder phenomena between the $M(6)$ and the $M(1, 2, 3)$ sites (which extensively occur in the presence of Fe^{2+}), and hence can provide unambiguous modelling of the effect of the B cations on the phase-transition behaviour.

FTIR spectroscopy can be used to detect the transition temperature on amphibole powders either from the merging of the two OH absorption bands observed in the $P2_1/m$ symmetry into the unique band observed in the $C2/m$ symmetry or from careful analysis of the spectrum in terms of peak shift and broadening in the MIR region (*e.g.*, Boffa Ballaran *et al.*, 2004). Both these methods require extrapolation of the data. In the case of two samples, we were able to test the accuracy of the FTIR-based approach by comparing the extrapolated T_c values with those obtained by single-crystal and powder X-ray diffraction (sample 334: Cámara *et al.*, 2003a; sample 407: Iezzi *et al.*, 2005, respectively).

Experimental methods

Details on the synthesis and characterization of the amphibole compositions examined here (Table 1) are reported in Iezzi *et al.* (2006). In brief, hydrothermal synthesis using silicate gels as the starting materials was done at 800 °C, 0.4 GPa using an internally heated vessel pressurized with Ar, at the University of Hannover (Germany). Selected-area electron diffraction (SAED) showed all samples to have $P2_1/m$ symmetry at room- T (Iezzi *et al.*, 2006).

FTIR spectra were collected on a Nicolet 760 spectrophotometer, equipped with KBr beamsplitter and a DTGS detector. The nominal resolution was 4 cm^{-1} and spectra are the average of 64 scans. Samples were prepared as KBr pellets; two disks were used, one for the OH-stretching region (5 mg of powder in 150 mg KBr) and a second for the MIR ($< 1300 \text{ cm}^{-1}$) region (1 mg of powder in 300 mg KBr). *In situ* high- T spectra were collected using a SpecacTM P/N 5850 high- T /high- P cell; the temperature was controlled using a Cr-Al thermocouple placed close to the sample and is accurate within ± 1 °C, according to the manufacturer. High- T spectra were collected with a step of 20 °C which was reduced to 10 °C when approaching the T_c predicted on the basis of the work of Cámara *et al.* (2003a) and Iezzi *et al.* (2005); each sample was equilibrated for 15 min at any target T before the FTIR data collection.

Table 1. Sample labels and nominal composition for the amphiboles of this study.

Sample	nominal composition
403	Na (NaMg) Mg ₅ Si ₈ O ₂₂ (OH) ₂
405	Na (Na _{0.6} Li _{0.4} Mg ₁) Mg ₅ Si ₈ O ₂₂ (OH) ₂
406	Na (Na _{0.2} Li _{0.8} Mg ₁) Mg ₅ Si ₈ O ₂₂ (OH) ₂
407	Na (LiMg) Mg ₅ Si ₈ O ₂₂ (OH) ₂

Room- T OH-stretching FTIR spectra

The room- T OH-stretching infrared spectra of the studied samples have been already discussed by Iezzi *et al.* (2004, 2005, 2006). They consist of two main bands at ~ 3740 and 3715 cm^{-1} , which are assigned to two independent OH groups in the $P2_1/m$ structure, which interact with the strongly off-centred ${}^A\text{Na}$, as shown by the structure refinement (Iezzi *et al.*, 2004). Two minor absorptions in the spectra are centred at 3690 and 3670 cm^{-1} , respectively. These latter bands are assigned to OH groups adjacent to vacant A sites in the structure, and indicate a slight departure from the nominal composition (cumingtonite component, *i.e.* vacancy at the A site locally balanced by excess ${}^B\text{Mg}$ with respect to 1.0 apfu).

High- T FTIR spectra: the OH-stretching region

Figure 1 shows the complete set of spectra collected at increasing T on sample 405. The main bands at 3740 and 3715 cm^{-1} gradually merge into a single, broad and symmetric absorption due to the $P2_1/m \rightarrow C2/m$ phase transition. The same kind of behaviour was recorded for all samples, but the merge of the two bands occurs at different temperatures. FTIR analysis also confirms that the transition is reversible and not quenchable; when decreasing the temperature, the OH-spectrum immediately recovers its initial two-band pattern.

It has been widely shown (*e.g.*, Salje *et al.*, 2000; Carpenter & Boffa-Ballaran, 2001; Hertwech & Libowitzky, 2002 and references therein) that structural changes occurring during a phase transition lead to changes in the phonon spectra. Such changes include peak shift, peak splitting and peak broadening; this latter feature is particularly significant in high- T spectroscopy. As a consequence, the transition temperature cannot be straightforwardly extracted from the OH-spectra given in Fig. 1. With increasing T , we actually observe two simultaneous effects: an intrinsic broadening of the two bands due to a temperature effect, and their merging due to the $P2_1/m \rightarrow C2/m$ phase transition. The combination of these two effects prevents a reliable deconvolution of the spectra if the FWHM or the peak positions is not constrained during the fitting process. In our opinion, therefore, the T of complete merging of the two components can be best located by visual inspection of the spectra, as the point beyond which there is no major change (Fig. 2). We estimate that T_c can be bracketed in this way with an uncertainty of

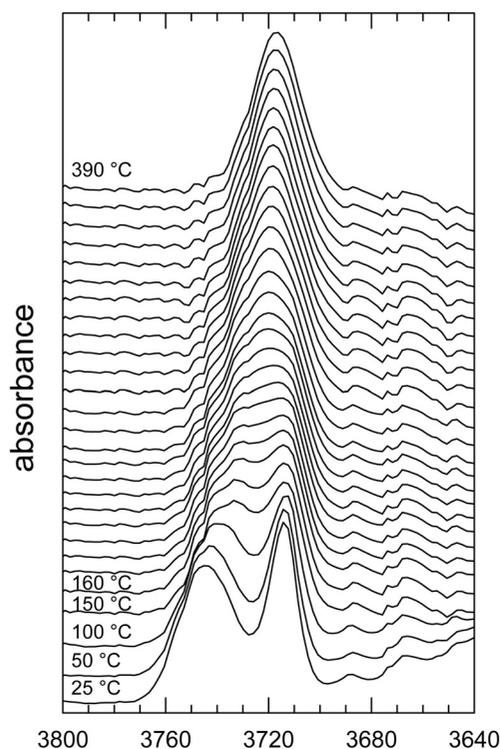


Fig. 1. Evolution of the OH-stretching region (sample 405) as a function of increasing T . From 160 °C up to 390 ° the T step is 10 °C.

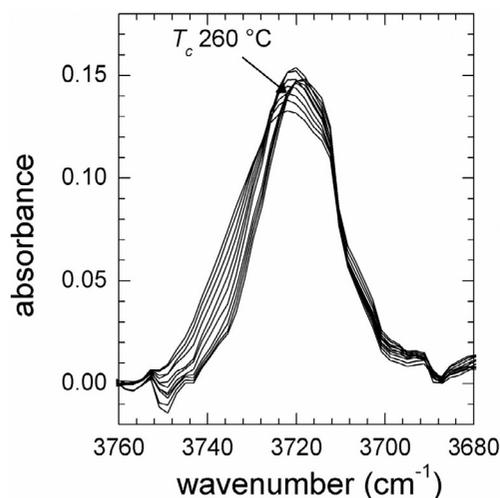


Fig. 2. Example of the “visual” estimation of the transition temperature using the OH-bands, sample 403. Note that after ~ 250 °C there are no more major changes in the spectrum.

1 ± 10 °C (because of the step used in collecting the data),
 2 which is however reasonable for comparison (see below).
 3 A list of the transition temperatures obtained by inspection
 4 of the OH spectra is given in Table 2. Note that these
 5 values represent the highest temperature beyond which no
 6 major change in the spectra is observed.

Table 2. Different estimates of T_c (in °C) obtained for the studied samples using the different methods, see text for explanation.

Sample	OH-region, visual estimate*	Peak shift of the A band**	Δcorr_{730} analysis
403	250(10)***	247(10)	230
405	280(10)	286(7)	270
406	320(10)	308(5)	310
407	340(10)	317(5)	320

* T beyond which there are no major changes in the spectra; ** fitting a 2-4-6 Landau potential to the variation of $\Delta\nu_A$ with T ; *** for this method the uncertainty can be considered to be ± 10 °C.

High- T FTIR spectra: the MIR (1300–640 cm^{-1}) region

Figure 3 shows the typical evolution of the FTIR spectrum in the medium infrared (MIR) 1300–640 cm^{-1} region as a function of increasing T . This spectral region is poorly studied for amphiboles and for silicates in general (see Milkey, 1960, and the classical book edited by Farmer, 1974). The many bands observed in the 1200–800 cm^{-1} range can be broadly assigned to the tetrahedral T-O stretching vibrations (Ishida, 1989, 1990); the multiplicity of components observed in Fig. 3 can be explained considering the different T-O bonds occurring in the amphibole structure. The bands in the 800–650 cm^{-1} range are usually assigned to the T-O-T and O-T-O stretching and/or bending vibrations, and the OH bending (δ_{OH}) modes (Ishida, 1989, 1990). Because from a structural point of view the $P2_1/m \rightarrow C2/m$ transition in amphiboles is mainly associated with geometrical changes in the double chains of tetrahedra (see Cámara *et al.*, 2003a for details), the spectra collected in the MIR region can be used to estimate the T_c , as already shown by Boffa Ballaran *et al.* (2001) in cummingtonite.

Figure 3 shows that the room- T spectrum in the 1300–820 cm^{-1} segment (segment 1) consists of overlapping bands centred at ~ 1150 (band A), 1100 (B), 1050 (C), 980 (D), 940 (E) and 915 (F) cm^{-1} , respectively. With increasing T , we observe a general broadening and a slight shift of all these bands; however, the most notable feature is the progressive merging of the two most intense components at 980 (D) and 940 (E) cm^{-1} .

At room T , the 820–640 cm^{-1} range (segment 2 in Fig. 3) also shows several absorptions: the two most prominent are centred at ~ 750 (G) and 670 (H) cm^{-1} , and a minor and broader absorption is observed at 720 (I) cm^{-1} ; all bands in this range simply shift and broaden with increasing T .

Peak shift as a function of T

As already shown to be the case for cummingtonite (Boffa Ballaran *et al.*, 2004), the parameter that can be determined with the highest precision in the spectral region of Fig. 3 is the wavenumber shift of the higher-frequency component A at 1140 cm^{-1} . However, the bands G and H, at ~ 750 and 670 cm^{-1} respectively, show the same type of shift.

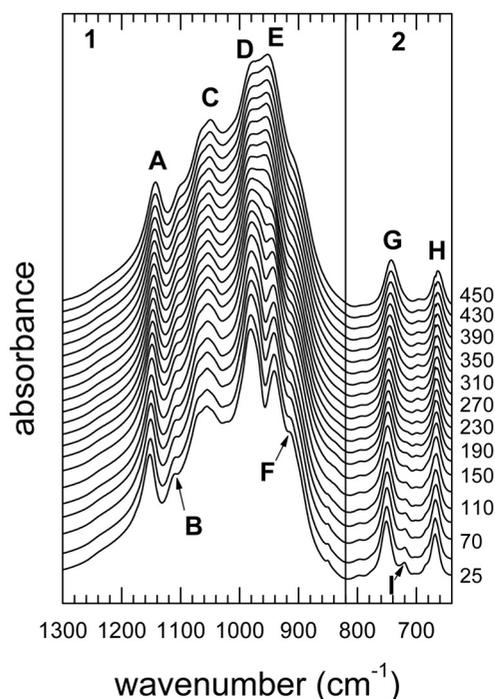


Fig. 3. Evolution of the FTIR spectra (sample 403) in the low-frequency 1300-640 cm^{-1} region as a function of increasing T ; temperatures given in $^{\circ}\text{C}$.

1 For all spectra, the exact peak position of band A was
 2 determined by analysing the second derivative of the in-
 3 tensity as a function of the wavenumber; the results are
 4 given in Fig. 4 as a function of T , and are compared
 5 with the analogous plot (Fig. 5) obtained for a synthetic
 6 richterite [$^{\text{A}}\text{Na}^{\text{B}}(\text{NaCa})^{\text{C}}\text{Mg}_5^{\text{T}}\text{Si}_8\text{O}_{22}^{\text{W}}(\text{OH})_2$], the amphibole
 7 with the closest composition and the same charge arrange-
 8 ment, which however does not undergo a $P2_1/m \rightarrow C2/m$
 9 phase transition. Comparison of Fig. 4 and 5 shows that,
 10 unlike richterite, all the LNMSH samples of this work are
 11 characterized by a well-defined break in the peak shift of
 12 the A band, therefore the data plotted in Fig. 4 can be used
 13 to determine the $P2_1/m \rightarrow C2/m$ transition temperature
 14 for these amphiboles. Linear extrapolation of the high- T
 15 data points to room- T provides the baseline from which
 16 one can calculate the excess frequency ($\Delta\nu_A$) at each tem-
 17 perature of the low- $T P2_1/m$ phase. Although not very large
 18 ($< 3.5 \text{ cm}^{-1}$), the values of $\Delta\nu_A$ obtained at room- T vary for
 19 each composition, with the largest excess value observed
 20 for the $^{\text{B}}(\text{LiMg})$ end-member. This feature agrees well with
 21 the larger spontaneous strain found by Iezzi *et al.* (2005)
 22 for sample 407 with respect to that determined by Cámara
 23 *et al.* (2003a) for a crystal with composition close to sam-
 24 ple 403.

25 The $\Delta\nu_A$ value is expected to vary with the square of a
 26 short-range-order parameter associated with the displacive
 27 phase transition, *i.e.* q^2 (Bismayer, 1990; Salje & Bismayer,
 28 1997). The variation of $\Delta\nu_A$ with T is not linear (Fig. 6),
 29 and can be fitted with a 2-4-6 Landau potential:

$$\Delta G = \frac{1}{2}a(T - T_c)q^2 + \frac{1}{4}bq^4 + \frac{1}{6}cq^6. \quad (1)$$

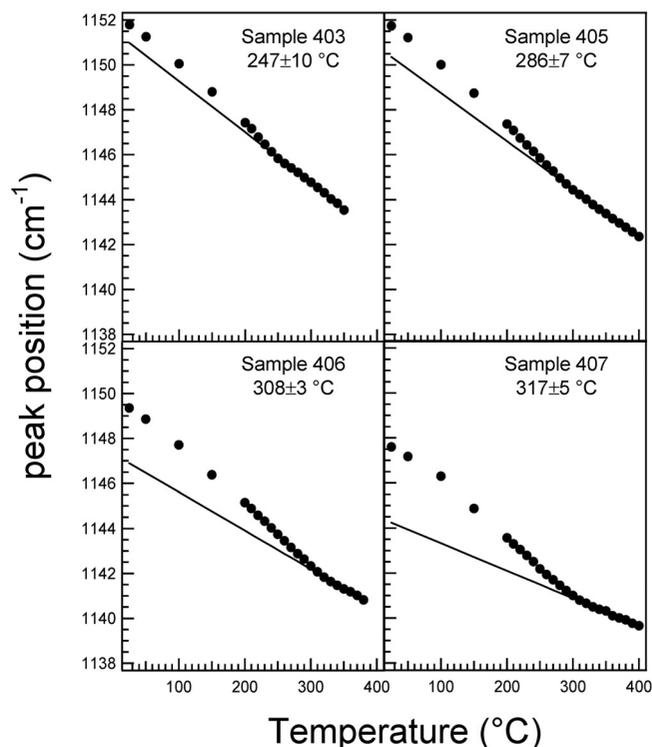


Fig. 4. Wavenumber shift of the high-frequency band A (see Fig. 3) observed as a function of increasing T for the studied samples. Lines are linear fitting to the high temperature data.

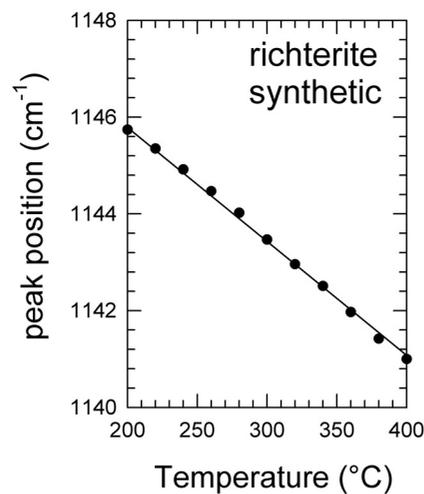


Fig. 5. Wavenumber shift as a function of T of the high-frequency A band for a richterite [$^{\text{A}}\text{Na}^{\text{B}}(\text{NaCa})^{\text{C}}\text{Mg}_5^{\text{T}}\text{Si}_8\text{O}_{22}^{\text{W}}(\text{OH})_2$] synthesized by Robert *et al.* (1989).

At equilibrium $\partial G/\partial(\Delta\nu_A) = 0$, and substituting q^2 with $\Delta\nu_A$, we can express the temperature as a function of $\Delta\nu_A$:

$$T = T_c - \frac{b}{a}\Delta\nu_A - \frac{c}{a}\Delta\nu_A^2. \quad (2)$$

The calculated transition temperatures are reported in Table 2, while the fitted ratios of Landau coefficients are reported in Table 3. Sample 407 has a c/a coefficient ratio

Table 3. Estimates of ratios of Landau coefficients obtained finding solutions to a 2-4-6 Landau potential.

Sample	b/a ($^{\circ}\text{C}$)	c/a ($^{\circ}\text{C}$)
403	-11 (67)	295 (80)
405	65 (25)	75 (17)
406	67 (7)	18 (5)
407	83 (9)	-1 (5)

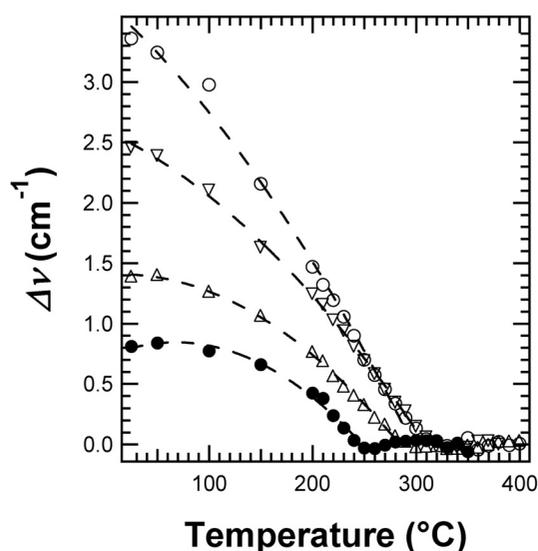


Fig. 6. Difference between the observed data and fitted baselines at high- T (see text for explanation) of the high-frequency A band for the studied compositions. Black dots = sample 403; upwards triangles = sample 405; downwards triangles = sample 406; open dots = sample 407. The dashed curves are 2-4-6 Landau potentials fits to data.

1 smaller than the error and is probably better fitted with a 2-
 2 4 potential (*i.e.* neglecting the q^6 term), although the results
 3 of such a fit compare well within errors ($T_c = 312(6)$ $^{\circ}\text{C}$
 4 and $b/a = 79(2)$ $^{\circ}\text{C}$). On the other hand, sample 403 has
 5 a negative and again not significant value for the b/a ratio
 6 (-11 with an estimated error of 67, *i.e.* five times larger)
 7 indicating tricritical behaviour. Fitting a 2-6 potential (*i.e.*
 8 neglecting the q^4 term) we obtain $T_c = 248(8)$ $^{\circ}\text{C}$ and
 9 $c/a = 282(20)$ $^{\circ}\text{C}$, again equivalent (within the error)
 10 to the results of the 2-4-6 fitting. Therefore, in order to keep
 11 internal consistency, all data have been fitted using a 2-4-6
 12 potential.

13 The same procedure has been applied to bands G and H in
 14 the low frequency region. The observed trends are similar
 15 to those of Figure 6, but the calculated $\Delta\nu_G$ and $\Delta\nu_H$ values
 16 are far below the nominal resolution of 4 cm^{-1} (< 1.6 cm^{-1}
 17 and < 0.5 cm^{-1} , respectively), and thus unsuitable for any
 18 reliable discussion.

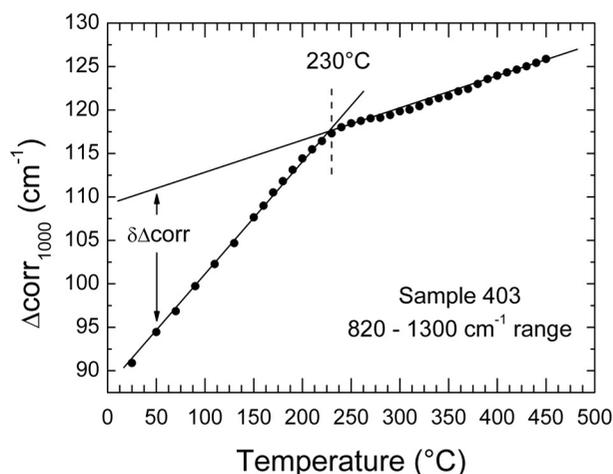


Fig. 7. Evolution of Δcorr_{1000} as a function of increasing T for sample 403.

proach has been widely used in crystal-chemical and thermo-
 23 dynamic studies during the last decade (see Malcherek
 24 *et al.*, 1995; Boffa Ballaran *et al.*, 1998, 2001; Hertwech &
 25 Libowitzky, 2002 among the others). The main advantage
 26 of this method is that it does not require any assignment of
 27 the observed absorptions to their phonon modes. An addi-
 28 tional advantage of this method is that it allows some
 29 insight into the thermodynamic character of the transition
 30 (*e.g.* Hertwech & Libowitzky, 2002).

31 For the samples of this work, we calculated Δ_{corr} val-
 32 ues (see Salje *et al.*, 2000 for a detailed description of the
 33 method) for the two spectral ranges 1300-820 cm^{-1} and
 34 820-640 cm^{-1} shown in Fig. 3.

35 Figure 7 shows the evolution with T of the Δcorr_{1000}
 36 calculated for sample 403 in the higher-frequency segment
 37 1 (820-1300 cm^{-1} : the intermediate frequency 1000 cm^{-1}
 38 being used as a label). The data show a well defined break
 39 at the transition temperature, and a linear fitting of the seg-
 40 ment above the transition yields $T_c = 230$ $^{\circ}\text{C}$. Similar evo-
 41 lutions, although not as well defined as that shown in Fig. 7,
 42 were obtained for all the other samples. The autocorrelation
 43 of the lower-frequency segment 2 (820-640 cm^{-1} , the in-
 44 termediate frequency 730 cm^{-1} being used as a label) gave
 45 more consistent results. An example of the evolution with
 46 T of the Δcorr_{730} for sample 406 is given in Fig. 8. Unfor-
 47 tunately, the observed evolution with T of the Δcorr val-
 48 ues does not allow a reliable fitting using a Landau poten-
 49 tial. Therefore, estimation of T_c has been done by finding
 50 the crossover point of the two lines obtained by fitting the
 51 high- T and low- T data. The T_c estimates obtained with this
 52 method are also given in Table 2, where comparison can be
 53 made with the values obtained using all the above described
 54 procedures.
 55

The $P2_1/m \leftrightarrow C2/m$ phase-transition in amphiboles as a function of T

All the LNMSH amphiboles have constant composition
 for the A, C and T cations and the O(5) anion; the only

19 Band broadening as a function of T

20 Changes in the line widths of the IR spectra occurring dur-
 21 ing a phase-transition can be conveniently modelled using
 22 the autocorrelation method (Salje *et al.*, 2000). This ap-

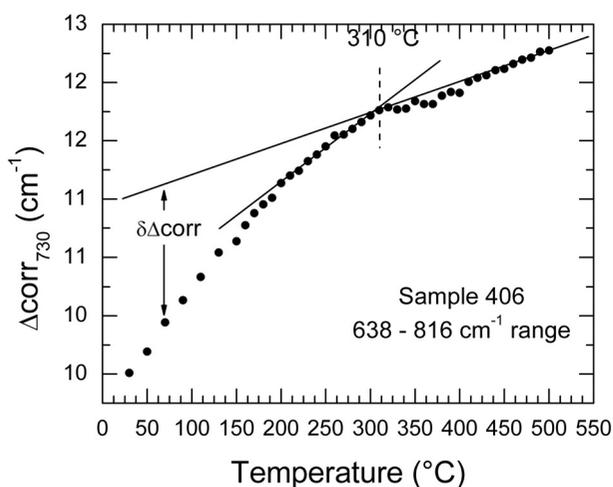


Fig. 8. Evolution of Δcorr_{730} as a function of increasing T for sample 406.

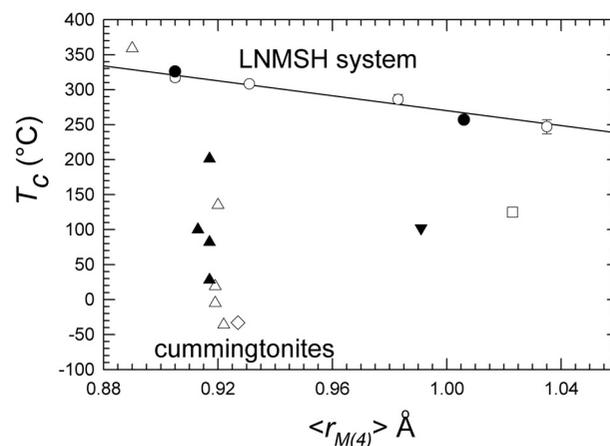


Fig. 9. Evolution of the transition temperature (T_c) as a function of the mean ionic radius of the $M(6)$ polyhedron ($\langle r_{M(6)} \rangle$). Open dots = this work; black dots = Cámara *et al.* (2003a) and Iezzi *et al.* (2005); filled downward triangle = Mn-rich parvo-tremolite (Reece *et al.*, 2000); open square = Cámara *et al.* (2008); black upward triangle = annealed samples from Boffa Ballaran *et al.* (2004); open upward triangle = natural samples from Boffa Ballaran *et al.* (2004); open diamond = Yang & Smith (1996). The line is a fit to the samples of the LNMSH system, and has the equation: T_c (°C) = 803–533 $\langle r_{M(6)} \rangle$ ($R^2 = 0.97$).

1 chemical variable in this system concerns the B cations
 2 and is represented by the ${}^B\text{Na}_1\text{Li}$ exchange (neglecting
 3 minor departures towards cummingtonite according to the
 4 ${}^A\text{Na}_{-1}\text{Na}_{-1}^A\text{Mg}_{-1}^B$ exchange).

5 Figure 9 compares the T_c calculated by fitting a Landau
 6 potential to the $\Delta\nu_A$ values for the studied samples with
 7 other T_c values from the literature plotted as a function of
 8 the aggregate ionic radius of the B cations (Shannon, 1976;
 9 given the different coordination of the various cations in
 10 the two symmetries, we used 8-fold coordination to ac-
 11 count for the whole local environment). In the LNMSH
 12 system, T_c correlates linearly with the aggregate radius of
 13 the B cations [T_c (°C) = 803–533 $\langle r_{M(6)} \rangle$; $R^2 = 0.97$]. The
 14 data obtained using X-ray diffraction on samples 334 and
 15 407 (Cámara *et al.*, 2003a; and Iezzi *et al.*, 2005; filled
 16 circle) fall on the same line. This behaviour is similar to
 17 what observed in cummingtonite, where the relevant ex-
 18 change is ${}^B\text{Mg}_2\text{Fe}_2$ (Yang & Hirschmann, 1995; Boffa
 19 Ballaran *et al.*, 2004); however the trend observed for these
 20 latter samples is significantly different. Examination of
 21 Fig. 9 shows that, although for both series T_c is positively
 22 correlated with $\langle r_{M(6)} \rangle$, the slope of the $P2_1/m \leftrightarrow C2/m$
 23 phase-boundary is different, being much steeper in cum-
 24 mingtonite than in LNMSH amphiboles. In other words,
 25 in cummingtonite a very small change in $\langle r_{M(6)} \rangle$ implies a
 26 greater change in T_c , and the $P2_1/m$ structure is thus far less
 27 stable than in the LNMSH system. Anyway, Fig. 9 confirms
 28 that T_c in amphibole is not a simple function of $\langle r_{M(6)} \rangle$, but
 29 that there are additional factors to consider to explain the
 30 strong difference between the two series.

31 In the cummingtonite series, where the A site is vacant,
 32 variations in B cations involve only small divalent cations
 33 (Mg, Fe, Mn) with relatively small differences in the ionic
 34 radii (0.89, 0.92 and 0.96 Å, respectively, in 8-fold coordi-
 35 nation). In contrast, in the synthetic series investigated
 36 here, the A site is almost filled by Na and both monovalent
 37 and divalent B cations are present in a 1:1 ratio; the range in
 38 the ionic radius of the B cations is much wider (Mg: 0.89,
 39 Li: 0.92 Na: 1.18 Å).

There are thus three possible factors to consider in order
 to understand why in these compounds the slope of the re-
 gression equation is less steep than in cummingtonite: (1)
 the A-site occupancy, (2) the presence of B cations with
 different charges and (5) the relative values of the aggre-
 gate radii of the B and C cations. Points (1) and (2) are
 obviously coupled because of local constraints related to
 bond-strength requirements, which force ordering of the A
 cation close to the monovalent B cation. Point (5) can be
 addressed by considering the work of Cámara *et al.* (2003a,
 2008), who measured a ΔT_c of + 132 °C due to the ex-
 change of F for OH in ${}^A\text{Na}^B(\text{NaMg})^C\text{Mg}_5^T\text{Si}_8\text{O}_{22}^W(\text{OH}, \text{F})_2$.
 This difference can only be ascribed to point (5), and corre-
 sponds to refined $\langle M-O \rangle$ distances of 2.077 and 2.065 Å
 for the OH and F end-members, respectively. Notably, the
 difference in $\langle M-O \rangle$ derives only from the shrinkage of
 the $M(1)$ and $M(5)$ sites [coordinated to F at O(5)]. No
 other data are available on fluorine end-members with dif-
 ferent mixtures of B cations; however, there is no reason to
 think that the trend should not be parallel to that of their
 OH-counterparts.

Point (5) is however also relevant in cummingtonite,
 where re-equilibration at higher T of a given composition
 allows Fe ordering at the $M(2)$ site at the expense of the
 $M(6)$ site (Yang & Hirschmann, 1995; Boffa Ballaran *et al.*,
 2001, 2002). Hence, $\langle r_{M(6)} \rangle$ decreases and $\langle r_{M(1,2,3)} \rangle$
 increases more rapidly, whereas T_c increases significantly.

Sueno *et al.* (1973) first proposed that, at least at the
 macroscopic scale, the driving force of the $P2_1/m \leftrightarrow C2/m$
 phase transition resides in the differential expansion of the
 various structure moduli. We want to suggest here that also
 the need for a local ordering of the A and monovalent B
 cations stabilizes the $P2_1/m$ structure: in fact, the A site can

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1 be more strongly off-centred in the $P2_1/m$ structure, where
 2 the non-equivalence of the two double-chains of tetrahedra
 3 also allows a better coordination for both small and large B
 4 cations.

5 The thermodynamic character 6 of the phase-transition

7 A quick inspection of the Landau coefficients reported in
 8 Table 3 shows that while the data for sample 407 can be
 9 fitted well with a 2–4 Landau potential (*i.e.* their c/a ratio
 10 is not significant), those of sample 403 are consistent
 11 with a strongly tricritical character (*i.e.* their b/a ratio is not
 12 significant); in the series, the calculated c/a value rapidly
 13 increases as a function of the ${}^B\text{Li}{}^B_1\text{Na}_1$ substitution.

14 Differently to the thermodynamic characters inferred by
 15 infrared spectroscopy, Cámara *et al.* (2003a, 2008) in their
 16 single-crystal diffraction experiments observed a 2nd order
 17 transition for both OH and F-rich ${}^B\text{Li}$ -free compositions,
 18 and Iezzi *et al.* (2005) observed a tricritical character
 19 for the (OH) ${}^B\text{Li}$ -rich composition (sample 407) in their
 20 powder-diffraction experiments. It is relevant to the present
 21 discussion to remind that diffraction experiments yield a
 22 long-range order parameter (Q), whereas FTIR analysis
 23 yields a short-range order parameter (q) which scales with
 24 $\Delta\nu_A$. Boffa Ballaran *et al.* (2004) already noted that the
 25 variation of the ratio I_b/I_a (the normalized sum of the intensity
 26 of selected superlattice reflections, which is $\propto Q^2$)
 27 was not collinear with $\Delta\nu_A$ ($\propto q^2$), whereas the T_c values
 28 obtained by both methods were in good agreement.
 29 The same conclusion can be drawn for the samples of this
 30 work, where the T_c values obtained by FTIR spectroscopy
 31 and diffraction methods are in a very good agreement. Our
 32 results thus provide further evidence that the local structure
 33 evolves in a different manner than the crystal structure
 34 which is averaged over thousands of unit cells.

35 Coupling between the aggregate radius 36 at the $M(6)$ site and the transition temperature

37 The Landau formalism can also be used to discuss how
 38 composition (and hence site-geometry) correlates (couples)
 39 with the occurrence of a displacive phase-transition. Boffa
 40 Ballaran *et al.* (2000), hypothesized that the value of the
 41 P_c can be renormalized by coupling of a order parameter
 42 related to the $P2_1/m \leftrightarrow C2/m$ phase-transition (Q)
 43 with composition ($X_{\text{Fe}}^{M(6)}$) and/or non convergent ordering
 44 (Q_{nco}), such as:

$$P_c^* = P_c - \frac{2\lambda_{\text{Fe}}X_{\text{Fe}}^{M(6)}}{a_v} - \frac{2\lambda_{\text{nco}}Q_{\text{nco}}}{a_v} \quad (3)$$

45 where, λ_{Fe} and λ_{nco} are the coupling coefficients and a_v is
 46 the Landau coefficient for the quadratic term; the dependence
 47 of the value of the critical temperature on X_{Fe} and/or
 48 Q_{nco} can be also worked out considering that Q_{nco} and Q

couple through common spontaneous strain (see Cámara
et al. 2003b for a detailed description) as:

$$T_c^* = T_c - \frac{2\lambda_{\text{Fe}}X_{\text{Fe}}^{M(6)}}{a} + \frac{2\lambda_{\text{s}}Q_{\text{nco}}}{a} \quad (4)$$

where a is the Landau coefficient of the quadratic term of
 the Landau expansion and λ_s is the common strain coupling
 coefficient. Equation (2) implies that increasing $X_{\text{Fe}}^{M(6)}$ (and
 consequently increasing $\langle r_{M(6)} \rangle$) reduces the value of the
 effective critical temperature (T_c^*). Cation disordering (*i.e.*
 the migration of Fe from $M(6)$ to $M(2)$, with a shortening
 of $\langle r_{M(6)} \rangle$) involves a decrease of Q_{nco} ; hence an increase
 of T_c^* is expected only if the sign of (λ_s/a) is negative. A
 negative sign has been indeed observed for the equivalent
 coupling coefficient in pigeonites by Cámara *et al.* (2003b).
 As a consequence, the determination of the sign of the coupling
 terms is essential when using equation (2) to quantify the
 coupling between composition and T_c . As shown in Fig. 9,
 there are different trends for Fe-Mg compositions with empty
 A sites, Mn-rich compositions with partially occupied A sites,
 F-rich compositions with filled A sites and OH-rich compositions
 with filled A sites. The compositional variation in the system
 studied here is simpler than in cummingtonites. As a matter of
 fact, we can renormalize T_c considering a compositional term
 related solely to the $M(6)$ site geometry, which is the only
 variable in this system. If we use $\langle r_{M(6)} \rangle$, T_c^* becomes:

$$T_c^* = T_c - \frac{2\lambda_{M(6)}\langle r_{M(6)} \rangle}{a} \quad (5)$$

The equation fitted above throughout the data points
 $[T_c$ (°C) = 803–533 $\langle r_{M(6)} \rangle$] implies that

$$\frac{2\lambda_{M(6)}}{a} = 533 \text{ (KÅ}^{-1}\text{)} \quad (6)$$

hence a significant coupling of the displacive transformation
 with $\langle r_{M(6)} \rangle$ is confirmed in the LNMSH system.

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